

### **In the Specification**

**Please amend the paragraph at Column 1, line 5, as follows:**

#### **TECHNICAL FIELD**

The present invention relates to an aromatic polyamide resin [moldings] molding having fine protrusions uniformly distributed over [their] its surface, particularly aromatic polyamide film that can be suitably used as a magnetic recording medium.

**Please amend the paragraph at Column 1, line 12, as follows:**

#### **BACKGROUND ART**

With high heat resistance and good mechanical properties, aromatic polyamides have various potential uses in the form of fiber, film, and aramid paper. Para-oriented aromatic polyamides, in particular, [are] have higher [in] rigidity, strength, and mechanical properties than other polymers, and are widely known as [the material] materials for high-strength fibers such as "Kevlar". When used as [material] materials for [film] films, they can produce very thin products, with potential [ones] uses including printer ink ribbon, magnetic tape, and capacitors. Aromatic polyamides [with] having better surface properties including running performance and abrasiveness are now demanded as more rapid processes for aromatic polyamide processing are adopted. [such as] They are used for producing coated-type or metal thin magnetic layers in magnetic recording [medium] media and forming image transfer layers for thermal recording. As magnetic recording medium products have been decreased in size and increased in recording density, formation of uniform, fine protrusions over the surface of aromatic polyamide film, which acts as the base film, is required to ensure a good balance among the output characteristics, running properties, and durability of the resulting magnetic recording medium. In the recently-developed magnetic recording medium consisting of a metal thin

magnetic layer, in particular, the surface characteristics of the base film can have a large influence on the surface characteristics of the magnetic layer, and this has increased the demands for such good base film materials.

**Please amend the paragraph at Column 1, line 40, as follows:**

Compared to aromatic polyimides and polyesters such as polyethylene terephthalate and polyethylene [naphthalate] naphthalate that have been conventionally used as base [film] films of magnetic recording [medium] media, aromatic polyamide film can be very thin due to its high rigidity and therefore is very suitable [material] for use in high-capacity magnetic recording [medium] media. The major difference from [the] polyester film which is produced by melt casting process is that an aromatic polyamide film is produced by a solution casting process. Compared to an aromatic polyimide film which is produced by solution casting process, aromatic polyamide film is also very different in [the] pH variations during the film forming process and in that polymerizing [solutions] solution contains inorganic salts in many cases, requiring particular care for conducting the desolvation step in the film forming process. Thus, special techniques suitable for aromatic polyamides have to be developed to solve these problems.

**Please amend the paragraph at Column 1, line 57, as follows:**

The use of inorganic particles contained in aromatic polyamide film is known as a technique to produce fine protrusions over the surface (e.g., Japanese Laid-Open Patent Applications (Kokai) [SH0] SHO60-127523 and SHO60-201914). Another known example is [a] an aromatic polyamide film consisting of a base layer laminated with a thin layer containing particles for producing surface protrusions (eg., Japanese Laid-Open Patent Applications (Kokai) HEI3-119512).

**Please amend the paragraph at Column 2, line 10, as follows:**

Second, finer protrusions have been increasingly required in recent years, as described above, and film containing smaller-diameter particles [is tend] tends to be used to meet this requirement. However, as the diameter of the particles [contained] decreases, the repulsion among particles decreases, tending to cause their coagulation, resulting in [a] decreased uniformity of [the] distribution of protrusions and an increased number of large protrusions. This is an essential [problems] problem arising from [then] the mixing of aromatic polyamide with particles of a foreign nature, and this problem cannot be avoided completely when solid particles, organic or inorganic, are used. Furthermore, aromatic polyamide during the process for preparing the polymer solution is liable to suffer large changes in pH of the solution to accelerate the coagulation of particles. Thus, the use of aromatic polyamide is disadvantageous regarding this particle coagulation.

**Please amend the paragraph at Column 2, line 26, as follows:**

Third, during such a processing process as stated above or during the use of a final product, aromatic polyamide film may often [travels] travel along guide rolls or guide pins, but the protrusions consisting of particles [contained] are generally so high in hardness that the surface of the guide will be scraped to produce dust which in turn will fall on the film. This problem will be particularly serious if the guide is made of a [plastics] plastic or if the film travels along it rapidly and repeatedly.

**Please amend the paragraph at Column 2, line 64, as follows:**

Japanese Laid-Open Application (Kokai) HEI 7-44857 has disclosed a film consisting of a magnetic surface with a special fine texture that is produced by blending two aromatic polyamide components, one being the major component and the other being the minor component different

from the former and lower in solubility, followed by a molding process in which the minor component aromatic polyamide is deposited on the surface. However, aromatic polyamides, particularly para-oriented ones, generally are not high in solubility, and if an aromatic polyamide is blended with another which is still lower in solubility, it will be difficult for them to mix completely to form a homogeneous polymer solution, even though complete dissolution seems apparent. A study by the present inventors has shown that film produced from such a polymer solution contains many rough protrusions with a height of 150 nm or more, despite the coexistence of fine protrusions, and that the surface tends to have crater-like dents that cause surface roughness. It has also been shown that if two aromatic polyamide components, major [Ad] and minor, that are very similar in structure and consequently nearly the same in solubility are blended, they seem to mix with each other completely and only a few protrusions are formed. Therefore, it is difficult for the technique disclosed in the Application to produce uniformly-distributed fine protrusions.

**Please amend the paragraph at Column 3, line 50, as follows:**

#### DISCLOSURE OF THE INVENTION

The present invention provides aromatic polyamide resin moldings, film in particular, at least one of whose surfaces [is] has 1.0 nm or more [in] root-mean-square roughness as observed by atomic force microscopy, and 80 nm or less [in] expressed as a ten-point mean, with a tensile Young's modulus of 9.8 GPa or more at least in one direction, and relates to production methods thereof, and magnetic recording [medium] media produced therefrom.

**Please amend the paragraph at Column 3, line 60, as follows:**

**BEST MODE FOR CARRYING OUT THE INVENTION**

Moldings of the present invention[s] that [are] have 1.0 nm or more [in] root-mean-square roughness (denoted as Rq hereinafter), as observed by atomic force microscopy as described later, and 80 nm or less [in] expressed as a ten-point mean (denoted as Rz hereinafter) can form uniform, fine protrusions when molded, and magnetic recording [medium] media, particularly those with a metal evaporated magnetic layer, produced from such moldings [will have] has small spacing loss with the head, and therefore, [will have] has good out put characteristics, and [will have] has good running properties and high durability due to [a] suitable roughness. An Rq value of less than 1.0 nm may lead to a loss of good slip characteristics, failure [inwindingup fi Imor] in winding up film or other products during their production, failure in conveying film during its processing, or reduced durability of magnetic recording medium produced. The Rq value should preferably be 2.0 nm [or more and] to 10 nm [or less], more preferably 2.5 nm [or more and] to 8.0 nm [or less]. An Rz of more than 80 nm may lead to a rough surface, defective coating in thermal copying material and magnetic recording medium, or a loss in the output characteristics of magnetic recording medium produced. The Rz value should preferably be 60 nm or less, more preferably 50 nm or less, further more preferably 40 nm. The lower limit is generally about 10 nm.

**Please amend the paragraph at Column 4, line 19, as follows:**

Moldings of the present invention [has] have a tensile Young's modulus of 9.8 GPa or more at least in one direction. If the Young's modulus is less than 9.8 GPa in [any] all direction, they may suffer a loss in processability or products may fail to have good properties. When used as a thin base for a magnetic recording medium, in particular, meeting the above-mentioned requirements is

important in [obtaining] making a high-output magnetic recording medium. The Young's modulus value should preferably be 11.7 GPa, more preferably 12.7 GPa, at least in one direction. Needless to say, the Young's modulus should preferably be 9.8 GPa in all directions. To meet these requirements, the para-oriented aromatic rings should account for preferably 50%, more preferably 75%, further more preferably 80%, still further more preferably 90%, of all aromatic rings contained in the aromatic polyamide to be used for the invention. Moreover, a dissimilar polymer as described later should be contained preferably up to less than 10 wt% of the total weight of the aromatic polyamide and [it] dissimilar polymer.

**Please amend the paragraph at Column 4, line 38, as follows:**

For the [surface] surfaces of moldings of the invention, the non-particle index for protrusions of 5 nm or more [high] should preferably be 80% or more, more preferably 85% or more, further more preferably 95% or more, still further more preferably 99% or more. Thus, surface protrusions as referred to in the present invention should preferably be formed through, for instance, phase separation between the aromatic polyamide and a dissimilar polymer as described below, instead of being formed by [using] adding particles. The aromatic polyamide and dissimilar polymer in a solution may be in a [complete] completely, or nearly [complete] completely, miscible state, but will form surface protrusions if phase separation is caused during the molding process. This virtually avoids the formation of large or non-uniform protrusions due to particle coagulation, which is an inherent problem with the addition of solid particles for protrusion formation. Since the aromatic polyamide and dissimilar polymer in a solution may be in a complete, or nearly complete, miscible state, as stated above, their mutual adhesion is strong after being molded, making removal of or damage to protrusions difficult [to occur] when an external force is applied.[, and] Further, the

hardness of the protrusions is lower than those formed by particle addition, protecting the rolls and guides from scraping during the manufacturing process or during the use of final [products] product.

**Please amend the paragraph at Column 4, line 63, as follows:**

For aromatic polyamide resin moldings of the present invention, the average height of the surface protrusions divided by their average diameter, hereinafter referred to as HD, should preferably be in the range from 1/40 to 1/2, more preferably 1/30 to 1/3, further more preferably 1/20 to 1/4. An HD value of[.] less than 1/40 may lead to a deterioration in the running properties of film products, while an HD value of more than 1/2 may lead to the formation of steep protrusions which may cause damage to rolls, or may be damaged themselves, during production or processing of film products etc.

**Please amend the paragraph at Column 5, line 7, as follows:**

The average diameter of such protrusions as described above should preferably be 30 nm [or more and] to 300 nm [or less], more preferably 50 nm or more and 200 nm or less, further more preferably 50 nm or more and 150 nm or less. The average diameter referred to herein is defined as the average of the size of each surface protrusion along its short and long axis. An average diameter less than 30 nm may lead to deformation of protrusions when an external force is applied during film production or processing, while material with an average diameter more than 300 nm may cause noise when used as support for a magnetic recording medium.

**Please amend the paragraph at Column 6, line 1, as follows:**

The number of protrusions of 15 nm or more in height should preferably be  $2 \times 10^4 / \text{mm}^2$  [mm2] or more, more preferably  $5 \times 10^4 / \text{mm}^2$  or more, further more preferably  $1 \times 10^5 / \text{mm}^2$  or more,

and the sum of the cross-section of each protrusion formed by a horizontal plane at a height of 15 nm should preferably account for 5% or less, more preferably 3% or less, further more preferably in the range of 0.001 to 3%, of the total cross-section to ensure further enhanced electromagnetic [transducing] conversion characteristics and durability.

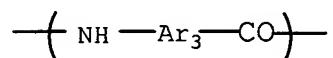
**Please amend the paragraph at Column 6, line 14, as follows:**

First, regarding the aromatic polyamide, the aromatic polyamide component of aromatic polyamide resin moldings of the invention should have a repeating unit as represented by general formula (I) and/or general formula (II) described below.

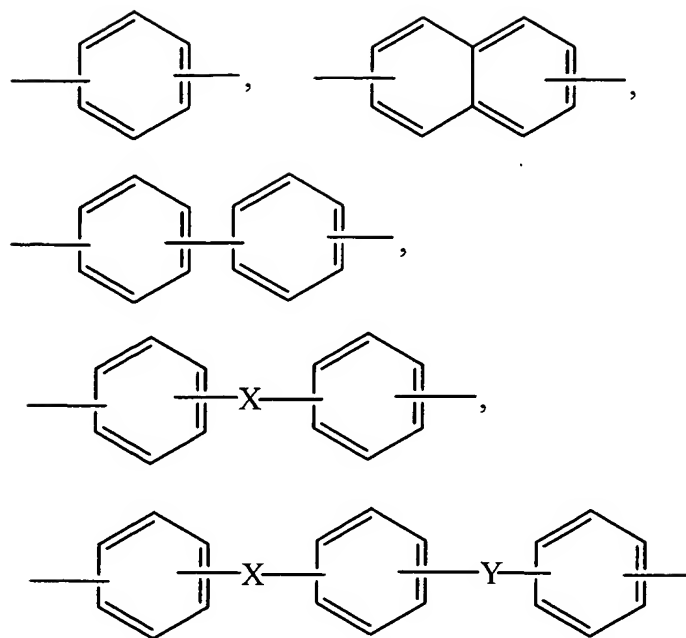
General Formula (I)



General formula (II)



where Ar<sub>1</sub>, Ar<sub>2</sub>, and Ar<sub>3</sub> may be, for instance,:



and X and Y may be selected from, but not limited to, the group of -O-, -CH<sub>2</sub>-, -CO-, -SO<sub>2</sub>-, -S-, and -C(CH<sub>3</sub>)<sub>2</sub>-. One or more of the hydrogen atoms on their aromatic ring may be replaced with a halogen atom such as fluorine, chlorine, and bromine (preferably chlorine), an alkyl group such as methyl, ethyl, and propyl (preferably methyl), or an alkoxy group such as methoxy, ethoxy, propoxy, and isopropoxy. In addition, one or more hydrogen atoms in the amide bond that forms the polymer may be replaced by a substituent. For enhanced characteristics, the above-mentioned aromatic rings that are bonded in para-orientation should preferably account for 50% or more, more preferably 75% or more, further more preferably 80% or more, still further more preferably 90% or more, of the total aromatic rings that are contained in the polymer to provide film with a high rigidity and high heat resistance. Para-orientation referred to herein is defined as the state where the bivalent bonds on the aromatic nuclei in the backbone chain are aligned parallel [a] to each other or [coaxially] coaxially. It is preferred that 30% or more of the total aromatic rings is accounted for by those in which one or more of the hydrogen atoms on the aromatic rings has been replaced with a halogen atom (preferably chlorine), as they will have enhanced characteristics in terms of increased moisture

resistance, reduced dimensional fluctuation caused by moisture absorption, and will prevent the rigidity reduction.

**Please amend the paragraph at Column 7, line 8, as follows:**

A dissimilar polymer as referred to herein is a polymer consisting of repeating units that are different from those in the above aromatic polyamide. One or more of such polymers may be used. They should preferably be [contained] present up to 0.1 wt% or more and less than 10 wt% relative to the total weight of aromatic polyamide and dissimilar polymer(s), as virtually uniform, fine protrusions will be formed in great numbers over the surface of the moldings. If their content is less than 0.1 wt%, protrusions will not be formed or, if formed, will be very small in number and height. If their content is more than 10 wt%, the domain of the dissimilar polymers formed through phase separation will be so large that the resultant surface protrusions will be too large in height and diameter to meet the requirements of the present [invent ion] invention. In addition, a content more than 10 wt% will allow the dissimilar polymers to occupy a large area, resulting in peeling at the interface with the aromatic polyamide that may reduce the toughness of the aromatic polyamide resin [A] moldings or cause a deterioration of the high rigidity and high heat resistance characteristic of aromatic polyamide. The content of dissimilar polymers should be appropriately fixed depending on [then] the type, solubility, molecular weight, etc., of the aromatic polyamide and dissimilar polymers used, and the size of the target moldings, etc., but it should preferably be 0.5 wt% or more and 8 wt% or less, more preferably 1 wt% or more and 6 wt% or less.

**Please amend the paragraph at Column 7, line 35, as follows:**

The types of such dissimilar polymers to be used should be selected depending on the design of the target surface, and there are no special limits on them. To meet the objectives of the invention,

however, it is preferred that they meet the following formulae where  $\delta_a$  is the solubility parameter of aromatic polyamide and  $\delta_b$  is the solubility parameter of the dissimilar polymer contained.

$$50(\text{MJ/m}^3)^{1/2} \leq \delta_a \leq 70(\text{MJ/m}^3)^{1/2}$$

$$2(\text{MJ/m}^3)^{1/2} \leq |\delta_a - \delta_b| \leq 20(\text{MJ/m}^3)^{1/2}$$

The solubility parameter used herein is calculated from the method proposed by Fedors (calculation process is shown, for instance, in Properties of Polymers, chapter 7, written by D. W. VanKreveren, 1976, Elsevier). For aromatic polyamides and dissimilar polymers of some special structures, the parameter cannot be calculated by the Fedors' method because the parameter concerning a chemical species contained is not known. In such a case, the parameter of a similar chemical species is used (for instance, the parameter for  $[-\text{SO}^2-]$   $-\text{SO}_2-$  is not known, so the parameters for  $-\text{S}-$ ,  $-\text{O}-$ , and  $-\text{O}-$  are used instead). The solubility parameter roughly indicate the [in] relative compatibility among different dissimilar polymers, and if  $\delta_a$  and  $\delta_b$  meet the above requirements, the size of the dispersed phase is controlled to allow the surface protrusions to meet the requirements of the present invention favorably. The value of  $|\delta_a - \delta_b|$  should preferably meet the following formula:

$$2[(\text{J/m}^3)] \underline{(\text{MJ/m}^3)}^{1/2} \leq |\delta_a - \delta_b| \leq 16(\text{MJ/m}^3)^{1/2}$$

more preferably,

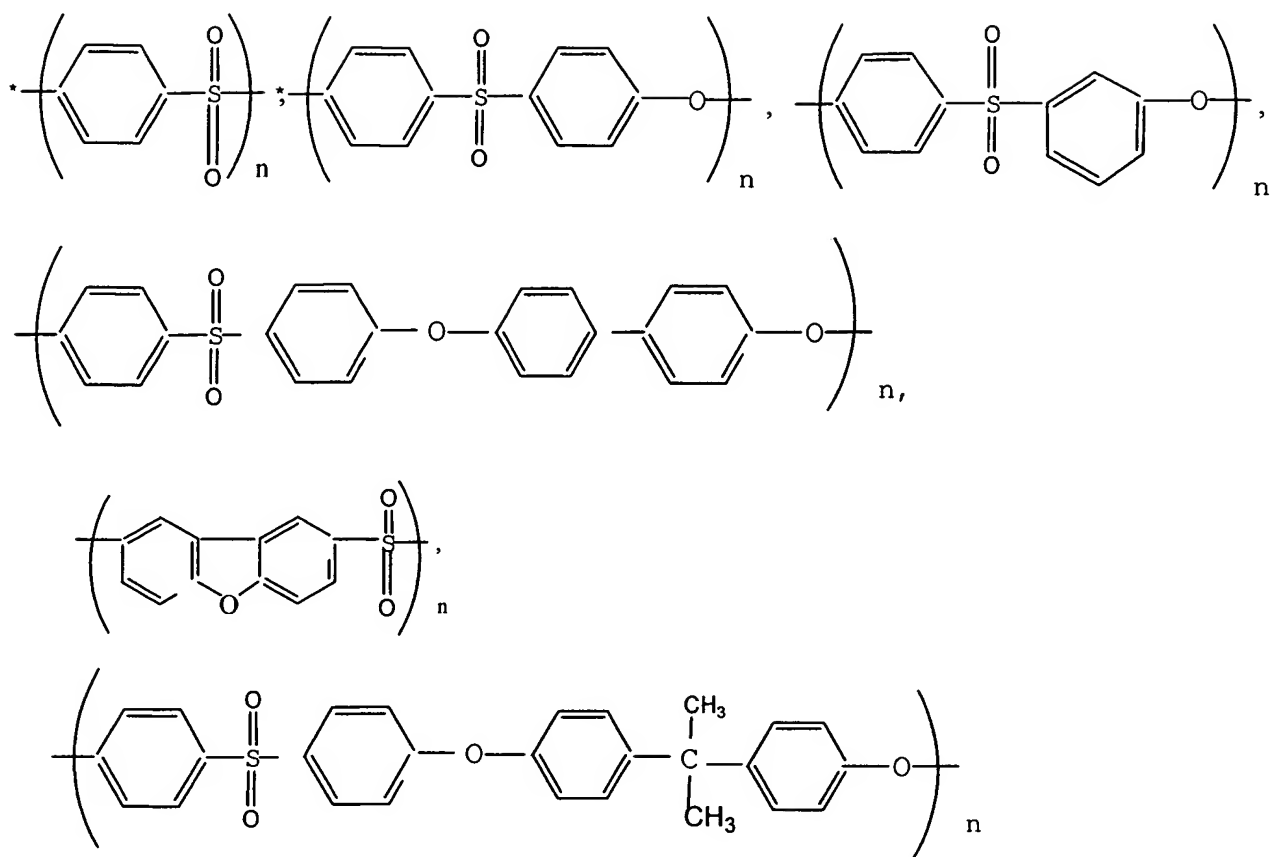
$$2(\text{MJ/m}^3)^{1/2} \leq |\delta_a - \delta_b| \leq 12(\text{MJ/m}^3)^{1/2}$$

**Please amend the paragraph at Column 8, line 8, as follows:**

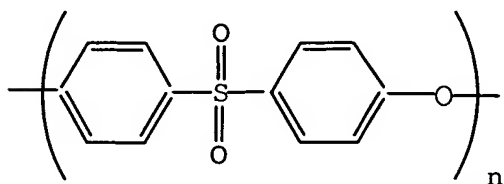
Such dissimilar polymers include polysulfone, polyethersulfone, polysulfide sulfone, polyphenylene sulfide, polyetherimide, [polyphenyleneoxide] polyphenylene oxide, [modifiedpolyphenyleneoxide] modified polyphenylene oxide, polyether ketone, polyetherether ketone, polycarbonate, polyethylene terephthalate, polybutylene terephthalate, and polyimide, and their precursor including polyamic acid, polyvinylidene fluoride, polymethyl methacrylate,

polystyrene, and polyvinyl alcohol. From the standpoint of [the] uniformity and heat resistance of the [surface protrusions] surface protrusions to be formed, it is preferred that at least one of the following polymers are [contained] present: aromatic polysulfon polymers including polysulfone, polyethersulfone, and polysulfide sulfone; aromatic polyetherimide polymers; polyphenylene oxide polymers including polyphenylene oxide, and modified polyphenylene oxide; aromatic polyketone polymers including polyether ketone, and polyetherether ketone; polycarbonate polymers; aromatic polyester polymers including polyethylene terephthalate, and polybutylene terephthalate; aromatic polyimide polymers including polyimide and those formed from polyamic acid, precursor for polyimide; of which aromatic polysulfon polymers are preferred most. The aromatic polysulfon polymers referred to herein are those which contain at least one sulfone group,  $\text{-SO}_2\text{-}$  in the repeating unit, such as polysulfone produced by condensation polymerization of sodium salt of bisphenol A with 4,4'-dichlorodiphenyl sulfone, or condensation polymerization of potassium salt of 4-(4-chlorophenyl sulfonyl) phenol. Specifically, they include known aromatic [iolsulfon] polysulfon polymers that consist of repeating units represented by formulae (III), which may be used alone or in combination.

Formulae (III)



where n denotes a positive integer, which should preferably be 5 or more and 1000 or less to ensure high heat resistance and solubility [a] to organic solvents. Of the structures given above, the one shown below is preferred to produce uniform protrusions:



**Please amend the paragraph at Column 9, line 20, as follows:**

Aromatic polysulfone polymers as listed above should preferably be [contained] present up to 0.1% or more and 10% or less, more preferably 0.5% or more and 8% or less, further more preferably 1% or more and 6% or less. Aromatic polysulfone polymers are generally not miscible with aromatic [polyamide] polyamides, but the present inventors have found that in a solution, the polymer can be highly compatible with aromatic polysulfone polymers if the latter are used in small amounts, and that uniform protrusions are produced through phase separation between the aromatic polyamide and aromatic polysulfone polymers if film is produced carefully from this solution as described below. If the content is less than 0.1wt%, resultant protrusions will not be sufficient in height and number, and slip properties will be poor, possibly leading to a reduced durability. If it is 10% or more, large protrusions will increase in number, and not only the resultant magnetic recording medium will be poor in electromagnetic conversion [propeprties] properties, but also film's mechanical properties may suffer deterioration.

**Please amend the paragraph at Column 9, line 39, as follows:**

Concerning the method for blending the aromatic polyamide and dissimilar polymer, the dissimilar polymer in the form of [Pellets] pellets or powder [maybe] may be added, directly or after being dissolved in a solvent, before or after the polymerization of the aromatic polyamide, but it is preferred that the aromatic polyamide and the dissimilar polymer is dissolved in separate solvents which are then blended to provide liquid material for molding. The solvents to be used for dissolving the aromatic polyamide and dissimilar polymer may be different from each other, but they should preferably be the same from the standpoint of industrial advantages including cost and productivity. Useful solvents include such organic solvents as N-methyl-2-pyrrolidone, dimethylacetamide,

dimethylformamide, hexamethylenephosphoramidate, dimethylimidazolidinon, and dimethylsulfonyl, as well as such mineral acids as concentrated sulphuric acid.

**Please amend the paragraph at Column 9, line 55, as follows:**

Aromatic polyamide resin moldings of the present invention are suitably produced by [blendingaromaticpolyamide] blending an aromatic polyamide and a dissimilar [polymers] polymer as described above, uniform, fine surface protrusions are formed without adding particles. Depending on their uses, however, the product may contain particles up to 0.0001 to 1.0 wt%. In such cases, the content should preferably be adjusted so that the non-particle index is 80% or more.

**Please amend the paragraph at Column 10, line 27, as follows:**

Depending on the uses of the product, the number of the above-mentioned particle protrusions on the surface should preferably be  $0.1 \times 10^4/\text{mm}^2$  [or more and] to  $20 \times 10^4/\text{mm}^2$  [or less], more [a] preferably  $0.3 \times 10^4/\text{mm}^2$  [or more and] to  $10 \times 10^4/\text{mm}^2$  [or less], further more preferably  $0.7 \times 10^4/\text{mm}^2$  [or more and] to  $2 \times 10^4/\text{mm}^2$  [or-less]. They may be preferably used as support for magnetic recording medium because they can improve the slip properties.

**Please amend the paragraph at Column 10, line 46, as follows:**

With their [excel lent] excellent surface properties, aromatic polyamide resin moldings of the present invention will be processed into such products as film and in-mold material, but film is most preferred to produce the intended effect of the invention. Studies by the present inventors have also shown that surface protrusions with very high uniformity are formed in the case of thin film of 1 to 20  $\mu\text{m}$ , preferably 2 to 10  $\mu\text{m}$  more preferably 2.5 to 7  $\mu\text{m}$ , in [height] thickness.

**Please amend the paragraph at Column 10, line 54, as follows:**

Film of the present invention has a tensile Young's modulus [Ha] of 9.8 GPa or more at least in one direction. This, combined with [excel lent] excellent surface properties, makes it possible to provide thin film products that have good processing properties and stability against external force despite their small thickness. In the case of magnetic recording medium, in particular, the output level of magnetic tape increases with the head-tape sliding properties, and therefore it is necessary to develop base film with a high Young's modulus. In particular, a high Young's modulus is required in the longer direction when a fixed head is used for recording, while it is required in the transverse direction when helical scanning is adopted. Base film [with] having a Young's modulus less than 9.8 GPa in [any direction] all directions is unfavorable because high output is not obtained regardless of the recording mechanism used. For aromatic polyamide film of the present invention, the Young's modulus should preferably be 11.7 GPa or more, more preferably 12.7 GPa, at least in one direction. Needless to say, it is preferred that the Young's modulus is 9.8 GPa or more in all directions.

**Please amend the paragraph at Column 11, line 7, as follows:**

[Film s] Film's elongation should preferably be 10% or more, more preferably 20% or more, further [morepreferably] more preferably 30% or more, at least in one direction to achieve a moderate degree of softness.

**Please amend the paragraph at Column 11, line 11, as follows:**

For magnetic recording, the film's moisture absorption coefficient should preferably be 5% or less, more preferably 3% or less, further more preferably 2% or less, to reduce the [tapers] tape's expansion and contraction caused by moisture changes, allowing good output properties to be maintained.

**Please amend the paragraph at Column 11, line 16, as follows:**

For magnetic recording, film's heat shrinkage at 200°C. for 10 minutes should preferably be 0.5% or less, more preferably 0.3% or less, to the [tapers] tape's expansion caused by temperature changes, allowing good output properties to be maintained.

**Please amend the paragraph at Column 11, line 21, as follows:**

Such film may be of a monolayer or a multilayer structure. To produce two-layered film, a solution of polymerized aromatic polyamide is divided into two parts, and two layers are built up after adding a dissimilar polymer to at least one part. The same process applies when film consisting of three or more layers is to be produced. Known methods to produce such a structure include the building up of layers in a die or in a feed block, and production of one layer followed by laminating it with another layer. In the case of a multilayer film, [a] the film surface of the present invention should preferably be formed as the outmost layer on at least one side of the film.

**Please amend the paragraph at Column 11, line 33, as follows:**

The present invention provides aromatic polyamide resin moldings from which film with good surface smoothness and running properties can be produced by controlling the height and number of protrusions on the target surface. It is particularly preferred that aromatic polyamide resin moldings of the present invention are used to produce film products, though they also can be processed favorably into fiber and in-mold material. Most preferred film products include external data storage [medium] media for computer and tapes for digital video-recording consisting of a thin metal magnetic layer that [is] are required to have [a surface] surfaces with high-density, highly uniform protrusions, though flexible printed circuit, capacitor, and printer's ink ribbon are also suitable uses.

**Please amend the paragraph at Column 11, line 53, as follows:**

There are no specific limitations [to] regarding the product shape, and the magnetic recording medium may be in the form of disk, card, or tape. As it is possible to produce very thin film with good surface properties and the high Young's modulus characteristic of the present invention[,]; a particularly preferred product is a long, high-density magnetic tape that consists of a support produced from film of the present invention that is 6.5  $\mu\text{m}$  or less in thickness, 2.3 to 13 mm in width, 100 m/roll or more in length, and 8 KB/mm<sup>2</sup> in magnetic recording density (uncompressed), because the surface profile can be [control led] controlled and [excel lent] excellent effects due to high rigidity can be further enhanced. The recording density referred to here is defined as the total recording capacity of a cassette divided by the area (length x width) of the magnetic tape. For magnetic recording medium, such as magnetic tape, there is increased demand recently for smaller, higher-capacity products, but there are important points as described below in producing high-capacity products. First, the thickness of the support should be reduced and the length of the product should be increased so that the total recording capacity is increased. And second, the track width should be reduced and shorter wavelength for recording should be used to increase the recording capacity per unit area. In most cases, both approaches tend to be used. In producing thin support, its material should be high in rigidity as a matter of course, and the surface of the support has greater contribution to the head-tape sliding properties and, in turn, the electromagnetic conversion properties, compared to a thicker support. When thick tape is used, the running tension and touch pressure of the head can be set to a high value, and therefore a stable contact can be maintained between the head and the tape even if the surface of the support is not [control led] controlled accurately. As the tape thickness becomes small, the running tension and touch pressure of the head have to be decreased, and therefore the head-tape contact and running properties become less uniform and less stable if the surface of the support is not controlled as described herein, possibly

resulting in head-track misalignment and loss of signals. An increased demand for increased data transfer rate has led to an increased relative velocity between head and tape, which increases the generation of frictional heat. To control it below a proper limit, the film of the present invention is very useful because the height and number of protrusions are [control led] controlled within a specific range. Thus, film of the present invention can provide magnetic tape products that effectively meet the requirement for higher capacity. The thickness of the support should preferably be 5.5  $\mu\text{m}$  or less, more preferably 4.5  $\mu\text{m}$  or less, and the recording density of the resultant magnetic recording medium should preferably be 25 KB/mm<sup>2</sup> or more, more preferably 34 KB/mm<sup>2</sup> or more.

**Please amend the paragraph at Column 12, line 36, as follows:**

Magnetic recording [medium] media of the present invention can be very useful for tape products for home use, professional use, broadcast use such as D-1, D-2, and D-3, digital video cassette, DDS-2, 3, and 4, data 8mm tape, QIC, and other data storage uses. [Most] The most preferred use is data storage in which greatest importance is placed on stability requirements related to loss of data, etc.

**Please amend the paragraph at Column 12, line 43, as follows:**

Many methods are available for producing the magnetic layer including coating methods in which magnetic powder such as iron oxide or powdered metal are kneaded with a thermosetting, thermoplastic, or radiation curable binder, followed by coating and drying, and by dry methods where thin magnetic metal layer is produced over the base film directly from Ni, Co, Cr, Fe,  $\gamma\text{-Fe}_2\text{O}_3$ , or other metals, or their alloys by[.] such a process as vapor deposition, sputtering, or ion plating. The use of a dry method is preferred because the resultant film can have very good surface properties characteristic of the present invention. When a dry method is adopted, the resultant magnetic

recording medium is further provided with a protective layer such as diamond-like coating, which may be further coated with a lubricant, in order to enhance the durability and slip properties.

**Please amend the paragraph at Column 12, line 62, as follows:**

Some production methods that can be used for the invention are described below, but this is not by way of limitation[s]. When aromatic polyamide is to be produced from an acid chloride and a diamine, solution polymerization may be performed in an [aprotonic] aprotic organic polar solvent such as N-methyl pyrrolidone (NMP), dimethylacetamide (DMAc), or dimethylformamide ([DUF]DMF), or interfacial polymerization may be performed in an aqueous medium. In such cases, care should be taken so that water or other substances that inhibit the reaction will not be allowed to come into the reaction system, in order to prevent the formation of low molecular weight components, and therefore, an effective stirring method should be used. The [material s] material's composition in terms of [equivalent] equivalents is important, but it should be adjusted properly in some cases to avoid deterioration of [the] film forming performance. Solubilizers such as calcium chloride, magnesium chloride, lithium chloride, lithium bromide, lithium nitrate may also be added.

**Please amend the paragraph at Column 13, line 13, as follows:**

Hydrogen chloride is produced as a byproduct if aromatic diacid chloride and aromatic diamine are used as monomers. Its neutralization may be performed with an inorganic neutralizer such as a salt consisting of a cation of a substance belonging to the first or second group of the periodic system such as calcium hydroxide, calcium carbonate, or lithium carbonate, and such an anion as hydroxide ion or carbonate ion, or an organic neutralizer such as ethylene oxide, propylene oxide, ammonia, triethylamine, triethanol amine, or diethanolamine. To improve the humidity-related properties of the base film, such substances as benzoyl chloride, phthalic anhydride, acetic

chloride, and aniline may be added to the system after the completion of the polymerization in order to cap the polymer ends. The reaction of isocyanate and carboxylic acid should be performed in [a] an aprotic organic polar solvent [under] in the [existence] presence of a proper catalyst.

**Please amend the paragraph at Column 13, line 30, as follows:**

Such a polymer solution as obtained may be blended with a dissimilar polymer, or the polymer may be separated and then dissolved in an organic solvent as listed above or in an inorganic solvent such as sulfuric acid to provide a stock solution to be used for the blending.

**Please amend the paragraph at Column 13, line 35, as follows:**

Organic or inorganic particles may be added during the above process.

**Please amend the paragraph at Column 13, line 37, as follows:**

To obtain an aromatic polyamide of the present invention, the intrinsic viscosity of the polymer (measured at 30°C. for a 100ml sulfuric acid solution containing 0.5g of the polymer) should preferably be 0.5 or more.

**Please amend the paragraph at Column 13, line 52, as follows:**

The stock solution as prepared above is filtered through a filter with a filtration accuracy of 6,000 nm or less, and then subjected to film production by a solution casting process. Solution methods include dry-wet, dry, and wet ones, but the use of a dry-wet or a dry method is preferred because surface protrusion formation by phase separation can be [control led] controlled easily in producing aromatic polyamide film of the present invention. If a dry-wet method is used for film production, the stock solution is extruded through a die onto support such as drum or endless belt to

form thin film, followed by drying so that solvents are removed from the film until the film [acquires] is self sustaining [property]. The drying conditions are one of the most important points in producing aromatic polyamide film of the present invention, and may have large influence on the surface properties of the resultant film. If the temperature of the support at the time of the casting of the stock solution,  $T_b$  (°C. ), and the temperature of the hot air introduced onto the cast film,  $T_a$  (°C. ), are in the range that meet the following formula, dissimilar polymers will be deposited all over the surface as a result of a convection current that is generated as the solvent evaporates, making it possible to effectively produce aromatic polyamide film with uniform, fine surface protrusions as proposed by the present invention.

$$20 \leq T_a - T_b$$

**Please amend the paragraph at Column 14, line 11, as follows:**

If  $T_a - T_b$  is less than 20°C, the temperature difference is not sufficiently large to generate a required convection current to produce sufficient protrusions. The difference should more preferably be 40°C. or more, further more preferably 50°C. or more. The upper limit of  $T_a - T_b$  is about 100°C, as long as excessive drying stain is produced. If an endless belt is used as support, a required temperature difference can be produced effectively during the polymer casting by adjusting the heating temperature above and below the endless belt or by cooling the film after the peeling of film. It is further preferred that the solvent is evaporated from the cast film which is then dried at a desolvating rate of 3-20%/min, more preferably 5-15%/min. If the desolvating rate is less than 3%/min, the protrusions may become so flat that they cannot meet the requirements of the present invention. If the desolvating rate is more than 20%/min, the number of large protrusions may become large [to] and increase the surface roughness, failing to meet the requirements of the present invention. Depending on the type of aromatic polyamide and/or

dissimilar polymers, the casting temperature at the die may be [control led] controlled within the range of 40°C. to 150°C. so that the aromatic polyamide polymer, which tend to suffer phase separation when in the form of a solution, is kept miscible in the solution whereas phase separation is caused at the time of molding, making it possible to produce film with surface protrusion as proposed in the present invention. The surface properties of the belt contact surface can be [control led] controlled by controlling the number of surface defects on the drum or endless belt used for the drying process. The number of surface defects with a diameter of 30  $\mu\text{m}$  or more should preferably be 0.001 to 0.02/mm<sup>2</sup>, more preferably 0.002 to [0.015/mm] 0.015/mm<sup>2</sup>.

**Please amend the paragraph at Column 14, line 61, as follows:**

The film is then led to a tenter for drying and/or heat treatment. The conditions for the drying and/or heat treatment are one of the most important points in producing aromatic polyamide film of the present invention. The drying and/or heat treatment is generally carried out by blowing hot air through a slit or circular nozzle against the film surface. To achieve surface properties of the present invention, it is preferred that the maximum temperature during this process is not less than the glass transition temperature of the dissimilar polymer (hereinafter denoted by  $T_g$  °C.) and not more than  $T_g + 100^\circ\text{C}$ ., because, if the maximum temperature is within this range, the dissimilar polymer tends to become spherical or elliptical due to its Brownian motion to meet the requirements of [t] the present invention. The speed of the hot air at the film surface [s h] should preferably be 1 to 30m/sec, more preferably 2 to 20m/sec, further more preferably 2 to 10m/sec, to effectively produce film of the present invention. If the air speed is less than 1m/sec, heat transfer in the film will not occur uniformly, heavily increasing the surface roughness in some cases. If the air speed is more than 30m/sec, crater-like irregularities may be produced, heavily increasing the surface roughness in some cases. As long as it is performed in a tenter, the drying and/or heat treatment at not more than the above-mentioned maximum temperature may be carried out at any point in the entire process. Heat treatment at the maximum temperature, for instance, may be performed after drying, or treatment at the maximum temperature may be followed by heat treatment at a lower temperature.

**Please amend the paragraph at Column 15, line 22, as follows:**

Film as produced as described above is stretched during the film production process so that the mechanical and thermal properties will meet the requirements of the present invention. The stretching ratio should preferably be 0.8 to 8.0, more preferably 1.1 to 5.0, in area (which is defined as the area of the film after stretching divided by that before stretching the ratio of 1 [II] indicating

relaxation). After this entire process, the resultant material is taken up by a winder to obtain aromatic polyamide film.

**Please amend the paragraph at Column 15, line 31, as follows:**

Slow cooling after the extension or heat treatment is effective. In particular, slow cooling at a rate of 50° C./sec or less [a] is effective.

**Please amend the paragraph at Column 16, line 19, as follows:**

(2) Non-Particle Index

The field emission-type scanning electron microscopy and energy dispersive X-ray analysis system described below (FE-SEM-XMA) is used to make analysis for not less than 100 protrusions selected arbitrarily and count the number of protrusions in which a particle-originating component is detected, followed by calculation of the non-particle index by the following formula:

Non-particle index (X) = 100 x (number of protrusions in which a particle-originating component is detected in the above measurement)/(measured number of protrusions)

Equipment: field emission-type scanning electron microscope S-800, manufactured by Hitachi, Ltd.

energy dispersive X-ray analyzer EMAX-3770, manufactured by Horiba, Ltd.

Measuring conditions

[accelerating] Accelerating voltage: 15 kV

[irradiation] Irradiation current: 0.2 nA

Measuring time: 100 sec

Preparation of specimen: carbon deposition onto specimen

**Please amend the paragraph at Column 16, line 40, as follows:**

### (3) Tensile Young's modulus and Elongation

A specimen 10 mm wide and 150 mm long, cut out of film, is extended in an Instron type tensile testing machine under the conditions of a chuck-to-chuck distance of 100 mm, stretching rate of 300 mm/min, chart speed of 500 mm/min, temperature of 23°C., and relative humidity of 65%. The tensile Young's modulus is determined from the tangent line to the rising part of the [weight-elongation] load-elongation curve obtained above. The elongation is determined from the length at break of the specimen minus the chuck-to-chuck distance, divided by the chuck-to-chuck distance and multiplied by 100.

**Please amend the paragraph at Column 16, line 65, as follows:**

#### 5) Film Dust/Particle Removal

A tape-like specimen (8 mm wide), [cutout] cut out from film, is allowed to travel on guide pins of stainless steel in a tape travel testing machine (TBT300D/H manufactured by Yokohama System Laboratory, [inc.] Inc.) under the conditions of a travelling speed of 200 m/min, back-and-forth travelling 10 times, winding angle of 180°, and travelling tensile stress of 1 kg/mm<sup>2</sup>. After the completion of travelling, a decision is made on the basis of the degree of film dust or particles found on the pins.

⊙: no dust or particles observed

○: small amounts in a few parts

Δ: small amounts in all parts

X: large amounts in all parts

**Please amend the paragraph at Column 17, line 12, as follows:**

#### (6) Resistance to Scratch

Continuous loading type scratch hardness testing machine HEIDON-18, manufactured by Shinto Chemical Co., is used to perform scratch hardness testing, and non-contact roughness analyzer TOPO-3D, manufactured by Wyco, [inc.] Inc., is used to determine the depth of flaws.

#### Measuring Conditions

scratching needle: tip curvature radius of 100  $\mu\text{m}$  (made of sapphire)

[weight: 0 100 g/100 mm] load: 0-100 g/100 mm

travelling speed: 10 m/min

#### Criteria

○: flaw depth less than 0.5  $\mu\text{m}$

$\Delta$ : flaw depth 0.5  $\mu\text{m}$  or more and less than 1.5  $\mu\text{m}$

X: flaw depth more than 1.5  $\mu\text{m}$

**Please amend the paragraph at Column 17, line 41, as follows:**

#### (8) Electromagnetic Conversion Properties

##### a) Initial Output Properties

Vacuum deposition is performed to form a magnetic layer on the surface of the film that is not in contact with the metal belt during the film [product ion] production process. A specimen 6.35 mm wide and 150 m long is cut off from the film, and set in a cassette. A sine wave is recorded with the optimum recording current, and the reproduction output from the film minus that from the reference tape is determined.

**Please amend the paragraph at Column 17, line 50, as follows:**

b) Dropout

The above tape cassette is placed on a video player, and signals of 4.4 MHz are recorded. The tape is then reproduced and the dropout at 15  $\mu$ sec-20dB is measured for 20 minutes with a dropout [USA] counter manufactured by Okura Industries, Ltd., followed by calculation of the dropout per minute.

**Please amend the paragraph at Column 17, line 57, as follows:**

(9) Durability

In an environment of 25°C. and 55 %RH, the tape produced in paragraph [4-a] 4a is allowed to run 100 times at 1,000 m/min on a 6 mm diameter guide pin with an angle,  $\theta$ , of  $\pi/2$  (rad) and tape-feed tension, T1, of 200 g, followed by measurement of the output and evaluation according to the following criteria.

o: difference from initial output less than 1 dB

$\Delta$ : difference from initial output 1 dB or more and less than 3 dB

X: difference from initial output more than 3 dB

Examples are given below, not by way of limitations, to further illustrate the present invention.

**Please amend the paragraph at Column 18, line 3, as follows:**

EXAMPLE 1

In N-methyl-2-pyrrolidone (hereinafter denoted by NMP), [2-chloroparaphenyldiamin] 2-chloroparaphenyldiamine, as an aromatic diamine component, and 4,4'-diamino diphenyl ether were dissolved up to a content equivalent to 80 mol % and 20 mol %, respectively, and 2-chloroterephthaloyl chloride was then added up to a content equivalent to 100 mol %, followed by stirring for 2 hours to complete the polymerization. The solution was neutralized with lithium

hydroxide to provide an aromatic polyamide solution with a polymer content of 10wt % and viscosity of 3,000 poise (hereinafter referred to as solution A).

**Please amend the paragraph at Column 18, line 46, as follows:**

**EXAMPLE 2**

Polyetherimide ("Ultem-1000", hereinafter PEI, manufactured by Japan GE Plastics) is dissolved in NMP up to 10 wt %, and blended with solution A prepared in Example 1 to provide a mixed solution in which PEI accounted for 1.5 wt % of the total amount of aromatic polyamide and PEI, followed by adequate mixing at [70° C] 70°C. for 3 hours.

**Please amend the paragraph at Column 19, line 12, as follows:**

The mixed solution thus obtained was filtered through filters with filtering accuracy of 5,000 nm and 1,000 nm, and, while being kept at a temperature of 50° C., cast onto an endless belt that had [2] surface defects of 30  $\mu\text{m}$  or more in diameter at a rate of 0.005/ $\text{mm}^2$ . At the time of casting, the temperature of the support,  $T_b$ , and the temperature of hot air,  $T_a$ , were 110° C. and 140° C., respectively. The desolvating rate was 7.2%/min. After being peeled off from the belt, the film was then immersed in a water bath at 40° C. for 5 minutes, dried at 160° C. for 30 seconds, and heat-treated in 280° C. hot air of a flow speed of 5 m/sec to produce aromatic polyamide film of 4.3 $\mu\text{m}$  in thickness. During the film production process, the film was stretched up to the ratio, 1.2 and 1.3 in the machine direction and the transverse direction, respectively.

**Please amend the paragraph at Column 19, line 29, as follows:**

**EXAMPLE 4**

Polycarbonate ("U-100", hereinafter PAR, manufactured by Unitika Ltd.) is dissolved in [NUP] NMP up to 10 wt %, and blended with solution A prepared in Example 1 to provide a mixed solution in which PAR accounted for 8.0 wt % of the total amount of aromatic polyamide and PAR, followed by adequate mixing at 50° C. for 3 hours.

**Please amend the paragraph at Column 19, line 37, as follows:**

The mixed solution thus obtained was filtered through filters with filtering accuracy of 5,000 nm and 1,000 nm, and, while being kept at a temperature of 50° C., cast onto an endless belt that had [2] surface defects of 30  $\mu\text{m}$  or more in diameter at a rate of [0.005/mm] 0.005/mm<sup>2</sup>. At the time of casting, the temperature of the support, Tb, and the temperature of hot air, Ta, were 140° C. and 170° C., respectively. The desolvating rate was 10.3%/min. After being peeled off from the belt, the film was then immersed in a water bath at 40° C. for 5 minutes, dried at 160° C. for 30 seconds, and heat-treated in 280° C. hot air of a flow speed of 5 m/sec to produce aromatic polyamide film of 4.3  $\mu\text{m}$  in thickness. During the film production process, the film was stretched up to the ratio, 1.2 and 1.3 in the machine direction and the transverse direction, respectively.

**Please amend the paragraph at Column 19, line 54, as follows:**

**EXAMPLE 5**

Paraphenyldiamine and terephthaloyl chloride were subjected to a known polymerization method, washed, and dried to produce polyparaphenylene terephthalamide (hereinafter referred as PPTA) with a  $\eta_{inh}$  [ $\eta_{inh}$ ] of 5.5 (intrinsic viscosity measured in concentrated sulfuric acid solution

with a polymer content of [0.5 g/10m] 0.5 g/10ml), which was dissolved in 99.5% concentrated sulfuric acid up to a polymer content of 12 wt %.

**Please amend the paragraph at Column 20, line 4, as follows:**

Powder of this polyamic acid was added to the PPTA solution prepared earlier so that the weight fraction of [Pi] PI becomes 4%, followed by stirring at 50° C. for 3 hours to ensure complete dissolution.

**Please amend the paragraph at Column 20, line 66, as follows:**

[Product ion] Production conditions and film properties are shown in Tables 1 to 5.

**Please amend Tables 1, 2, 3 and 5, as follows:**

TABLE 1

	Dissimilar polymer or particle	Content (wt %)	$\delta_a$ (MJ/m <sup>3</sup> ) <sup>1/2</sup>	$\delta_b$ (MJ/m <sup>3</sup> ) <sup>1/2</sup>	$\delta_a - \delta_b$ (MJ/m <sup>3</sup> ) <sup>1/2</sup>	Polymer temp. at [discharge] casting (° C.)	Hot air temp. Ta (° C.)	Support temp. [tb]Tb(° C.)
Example 1	PES	3	56.6	53.6	3.0	60	170	140
Example 2	PEI	1.5	56.6	55.7	0.9	120	170	140
Example 3	PC	2	56.6	49.2	7.4	50	140	110
Example 4	PAR	8	56.6	51.1	5.5	50	140	110
Example 5	PI (polyamic acid)	4	59.9	58.9	1.0	60	90 (70 RH %)	60
Example 6	PES	3	56.6	53.6	3.0	60	170	140
	colloidal silica	0.1						
Comparative example 1	none	none	56.6	-	-	60	170	140
Comparative example 2	PEI	12	56.6	55.7	0.9	120	170	140
Comparative example 3	[PFS] PES	0.3	56.6	53.6	3.0	60	170	170
Comparative example 4	colloidal silica	2	56.6	-	-	60	170	140

TABLE 2

	Desolvating rate (%/min)	Drying temp. (° C.)	Heat treatment temp. (° C.)	Heat treatment air speed (m/sec)	Stretching ratio in machine direction	Stretching ratio in transverse direction
Example 1	10.2	160	280	5	1.2	1.3
Example 2	10.2	160	280	5	1.2	1.3
Example 3	7.2	160	280	5	1.2	1.3
Example 4	10.3	160	280	5	1.2	1.3
Example 5	3.2	160	350	5	1.1	1.25
Example 6	9.9	160	280	5	1.2	1.3
Comparative example 1	10.1	160	280	5	1.2	1.3
Comparative example 2	10.3	160	280	5	1.2	1.3
Comparative example 3	20.5	160	280	5	1.2	1.3
Comparative example 4	10.5	160	280	5	1.2	1.3

[45] 5

TABLE 3

	Rq (nm)	Rz (nm)	Ra (nm)	Rt (nm)	Rt/Ra	Non-particle index (%)	HD [(nm)]	Average protrusion diameter (nm)
Example 1	2.9	23.2	1.9	28.0	14.7	100	0.07	190
Example 2	4.8	57.2	3.7	66.2	17.9	100	0.13	420
Example 3	1.5	12.0	0.9	13.5	15.0	100	0.06	60
Example 4	2.8	30.2	2.2	44.0	20.0	100	0.15	350
Example 5	3.1	28.4	2.3	32.4	14.1	100	0.10	210
Example 6	3.7	33.5	2.3	40.7	17.7	99	0.12	180
Comparative example 1	0.9	11.3	0.8	17.5	21.8	100	0.02	20
Comparative example 2	12.5	120	10.8	250	23.1	100	0.16	560
Comparative example 3	0.9	10.5	0.7	13.5	19.3	100	0.06	30
Comparative example 4	8.4	88.0	6.7	105	15.7	3	0.66	40

TABLE 5

	Resistance to scraping	Film dust/ particle removal	Resistance to scratch	Output (dB)	Dropout [number/min)] (number/min)	Durability
Example 1	◎	◎	○	2.3	0.3	○
Example 2	○	○	○	-0.3	1.1	△
Example 3	◎	○	△	2.0	1.5	△
Example 4	○	○	△	1.9	0.8	△
Example 5	◎	◎	○	2.0	0.5	○
Example 6	◎	◎	○	1.5	1.1	○
Comparative example 1	◎	X	X	2.7	0.3	X
Comparative example 2	X	X	△	-2.3	3.2	△
Comparative example 3	△	X	X	1.4	0.3	X
Comparative example 4	X	X	△	-1.3	3.7	X

**Please amend the paragraph at Column 23, line 55, as follows:**

Elsewhere, a 15 wt % amount of dried polyethersulfone PES-E2010 (hereinafter referred to as PES), manufactured by Mitsui Toatsu Chemicals, Inc., was dissolved in NMP, [and] and this PES solution was added to solution A to provide a mixed solution in which PES accounted for 3 wt % relative to aromatic polyamide. This solution is hereinafter referred to as solution B.

**Please amend the paragraph at Column 23, line 62, as follows:**

Solution B thus obtained was filtered through a 5 $\mu$ m-cut filter, cast onto a stainless steel endless belt with a mirror finished surface, and heated at 150° C. for 5 minutes to evaporate the solvent until the film acquired [self-sustain] self-sustaining property. The film was then peeled off from the belt continuously. At this point, the polymer content in the gel film was 39.8 wt %, and the desolvating rate was 5.9%/min. The film was then immersed in a water bath for 2 minutes for aqueous extraction of the remaining solvent and inorganic salts that had resulted from neutralization. During this process, the film was stretched up to the ratio of 1.2 in the machine direction. Subsequently, while being dried and heat-treated in 280° C. hot air of a flow speed of 3 m/sec, the film was stretched up to the ratio of 1.3 in the transverse direction. The film was further heat-treated at 250° C. for 1.5 minutes, followed by slow cooling at 20° C./sec to produce aromatic polyamide film of 4.2  $\mu$ m in thickness.

**Please amend the paragraph at Column 24, line 60, as follows:**

#### EXAMPLE 8

Spherical silica with an average particle diameter of 20 nm was added to polymer solution B prepared in Example [1] 7 up to 0.2 wt % relative to polyamide, followed by film production by the same procedure as in Example 7.

**Please amend Table 8 as follows:**

TABLE 8

	Rq (nm)	Rz (nm)	Ra (nm)	Rt (nm)	Rt/Ra	Non-particle index (%)	HD [(nm)]	Average protrusion diameter (nm)
Example 7	2.5	21.3	1.8	26.3	14.6	100	0.06	220
Example 8	2.7	27.5	1.9	30.4	16.0	99	0.06	210
Example 9	3.3	29.2	2.5	32.3	12.9	100	0.05	280
Example 10	4.1	33.6	3.2	44.0	13.8	100	0.11	330
Comparative example 5	0.9	11.0	0.8	16.3	20.4	100	0.03	30
Comparative example 6	8.4	81.5	8.1	97.0	12.0	100	0.10	380

**Please amend the paragraph at Column 27, line 33, as follows:**

**EXAMPLE 11**

Colloidal silica with a primary particle diameter of [30] 80 nm was dispersed adequately in NMP and added to aromatic polyamide up to 0.02 wt %, followed by polymerization by the same procedure as in Example 1. The resultant solution was blended with a PES solution as in Example 1 to produce a mixed solution, which was filtered through filters with filtering accuracy of 5,000 nm and 1,000 nm, and, while being kept at 60° C., cast onto an endless belt that had surface defects of 30 [cm] μm or more in diameter at a rate of 0.005/mm<sup>2</sup>. At the time of casting, the temperature of the support, Tb, and the temperature of hot air, Ta, were 120° C. and 150° C., respectively. The desolvating rate was 8.5 %/min. After being peeled off from the belt, the film was then immersed in a water bath at 40° C. for 5 minutes, dried at 160° C. for 30 seconds, and heat-treated in 250° C. hot air of a flow speed of 3 m/sec to produce aromatic polyamide film of 4.3 μm in thickness. During the film production process, the film was stretched up to the ratio, 1.2 and 1.3 in the machine direction and the transverse direction, respectively.

**Please amend the paragraph at Column 27, line 61, as follows:**

**EXAMPLE 12**

Polymerization was performed to produce a mixed solution by the same procedure as in Example 11 except that colloidal silica with a primary particle diameter of 50 nm was added up to 0.005 wt % relative to aromatic polyamide and that PES was added up to 6 wt % relative to aromatic polyamide. The mixed solution thus obtained was filtered through filters with filtering accuracy of 5,000 nm and 1,000 nm, and, while being kept at 60° C., cast onto an endless belt that had surface defects of 30 μm [cm] or more in diameter at a rate of 0.005/mm<sup>2</sup>. At the time of casting, the temperature of the support, Tb, and the temperature of hot air, Ta, were 120° C. and [150 ° C.] 170° C., respectively. The desolvating rate was [8.5%/min] 7.2 %/min. After being peeled off from the

belt, the film was then immersed in a water bath at 40° C. for 5 minutes, dried at 160° C. for 30 seconds, and heat-treated in 250° C. hot air of a flow speed of 6 m/sec to produce aromatic polyamide film of 4.3 µm in thickness. During the film production process, the film was stretched up to the ratio, 1.2 and 1.3 in the machine direction and the transverse direction, respectively.

**Please amend the paragraph at Column 28, line 51, as follows:**

Production conditions and film properties are shown in Tables [a] 11 to 15.

**Please amend the paragraph at Column 28, line 66, as follows:**

Production conditions and film properties are shown in Tables [10] 11 to 15.

**Please amend Tables 11 and 13 as follows:**

TABLE 11

	Dissimilar polymer or particle	Content (wt %)	$\delta_a$ [(MJ/mm <sup>3</sup> ) <sup>1/2</sup> ] (MJ/m <sup>3</sup> ) <sup>1/2</sup>	$\delta_b$ (MJ/m <sup>3</sup> ) <sup>1/2</sup>	$ \delta_a - \delta_b $ (MJ/m <sup>3</sup> ) <sup>1/2</sup>	[Content (wt %)] Polymer temp. at casting (° C.)	Hot air temp. Ta (° C.)	Support temp. Tb (° C.)
Example 11	PES	3	56.6	53.6	3.0	60	150	120
	colloidal silica	0.02						
Example 12	PES	6	56.6	53.6	3.0	60	170	120
	colloidal silica	0.005						
Comparative example 7	PES	3	56.6	53.6	3.0	60	170	140

TABLE 13

	Rq (nm)	Rz (nm)	Ra (nm)	Rt (nm)	Rt/Ra	Non-particle index (%)	HD[(nm)]	Average protrusion diameter (nm)
Example 11	1.7	18.5	1.2	17.8	14.8	99	0.03	70
Example 12	2.7	24.2	1.8	30.4	16.9	99	0.10	140
Comparative example 7	1.2	12.1	1.0	12.5	12.5	100	-	-

\* Comparative example 7: Heavy roughness due to crater-like dents was seen all over the surface.

**Please amend the paragraph at Column 31, line 2, as follows:**

**INDUSTRIAL APPLICABILITY**

The invention provides aromatic polyamide resin moldings whose surface have protrusions that are virtually uniform, fine, and so high in affinity with aromatic polyamide that they [are hard to be destroyed] resist destruction.

**Please amend the paragraph at Column 31, line 7, as follows:**

Further, the invention provides aromatic polyamide film that, when used as base film for a magnetic recording medium, are [excel lent] excellent in electromagnetic conversion properties and durability.